

## EDS User School

# Principles of Electron Beam Microanalysis



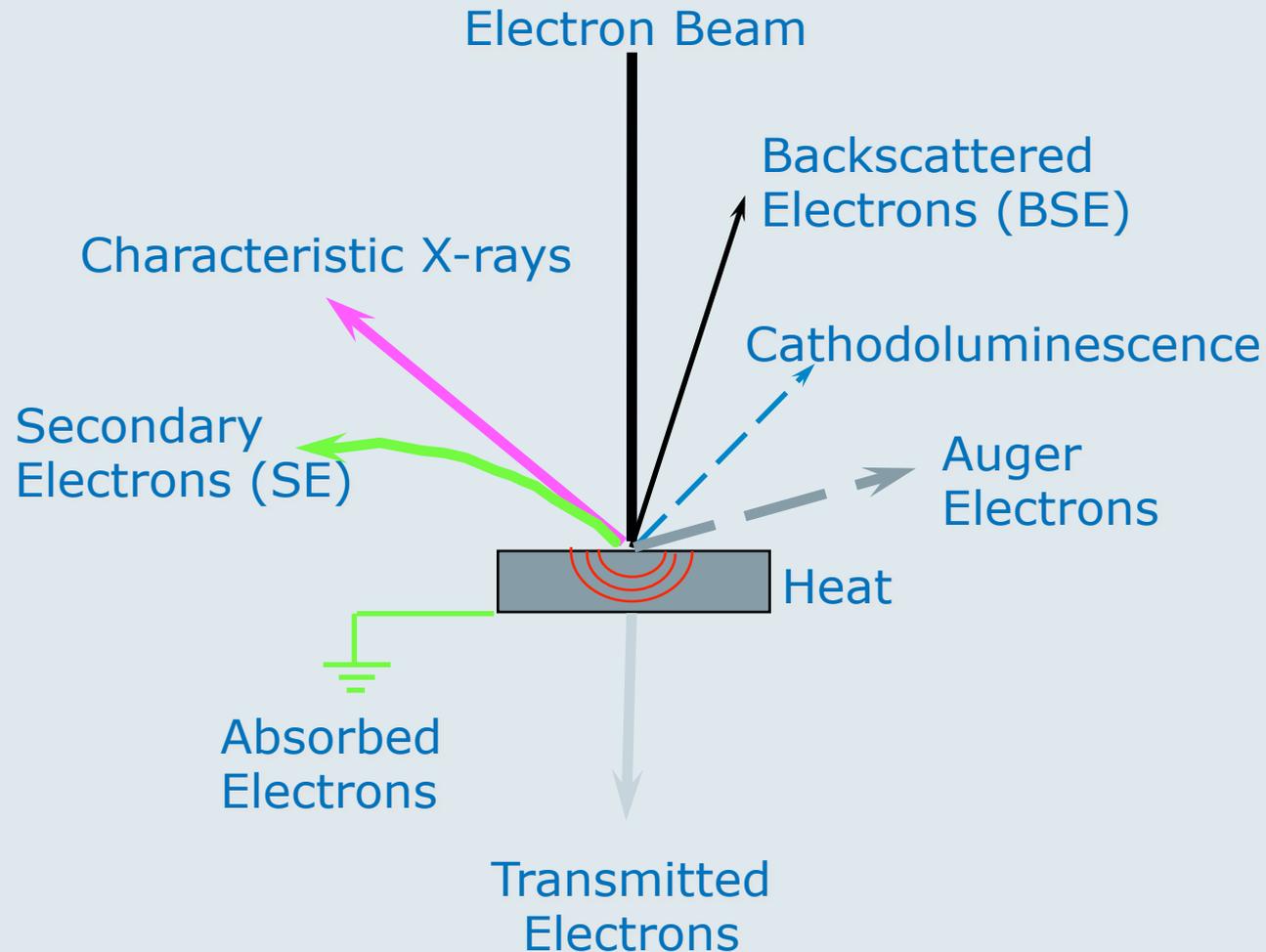
XFlash<sup>®</sup>  
EDS

H																	He	
Li	Be											B	C	N	O	F	Ne	
Na	Mg											Al	Si	P	S	Cl	Ar	
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr	
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe	
Cs	Ba	La	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn	
Fr	Ra	Ac																
			Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu		
			Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lw		

# Outline

- 1.) Beam-specimen interactions
- 2.) EDS spectra: Origin of Bremsstrahlung and characteristic peaks
- 3.) Moseley's law
- 4.) Characteristic peaks: K-, L-, and M series
- 5.) Spatial resolution and excitation range in EDS analysis

# 1.) Beam-specimen interactions



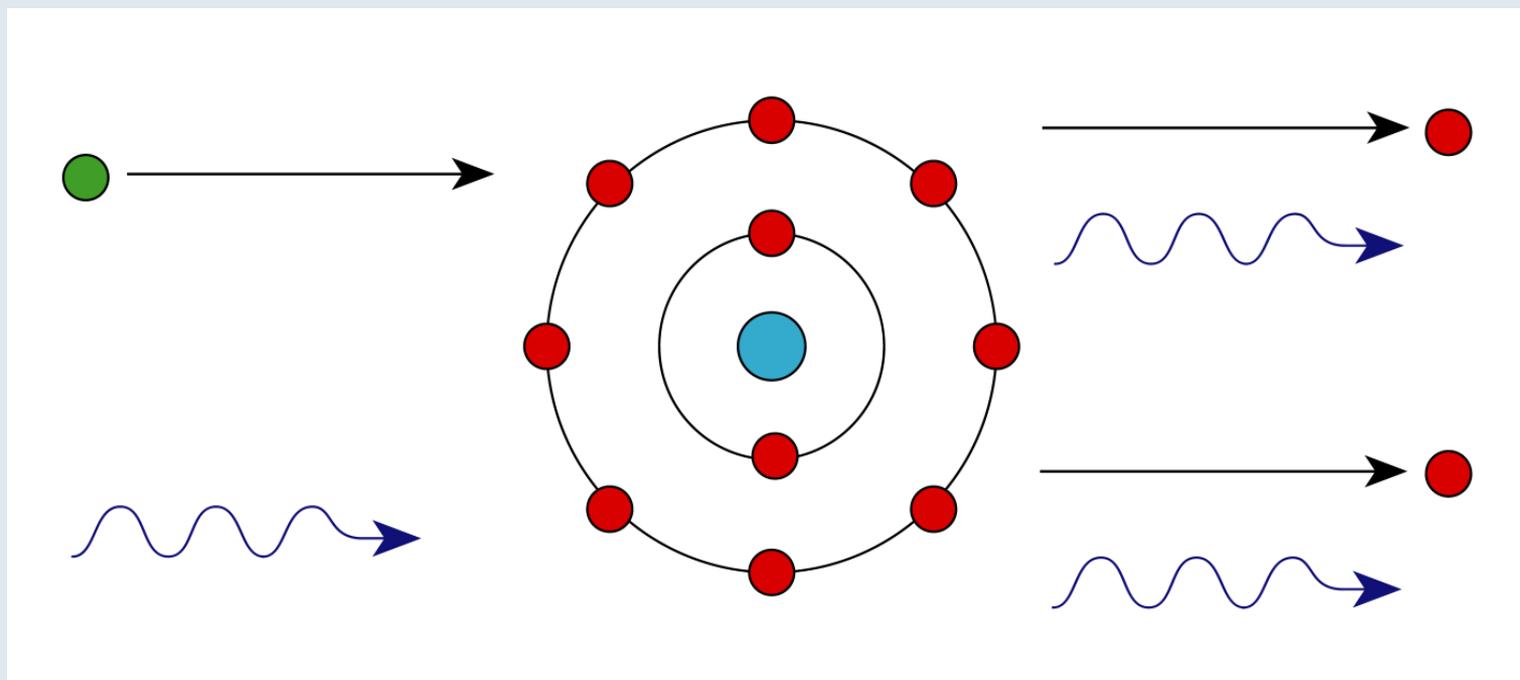
# 1.) Beam-specimen interactions

Accelerated from a source \ Emitted from the specimen	Electrons	Ions	X-rays	Light
Electrons	SEM, AES		XPS	UPS
Ions		SIMS		
X-rays	EDS, WDS	PIXE	XRF	
Light	CL			

SEM - scanning electron microscopy  
 WDS - wavelength dispersive spectrometry  
 XPS - X-ray photoelectron spectroscopy  
 SIMS - secondary ion mass spectrometry  
 PIXE - proton-induced X-ray emission

EDS - energy dispersive spectrometry  
 AES - Auger electron spectroscopy  
 UPS - UV-light photoelectron spectroscopy  
 XRF - X-ray fluorescence spectroscopy  
 CL - cathodoluminescence

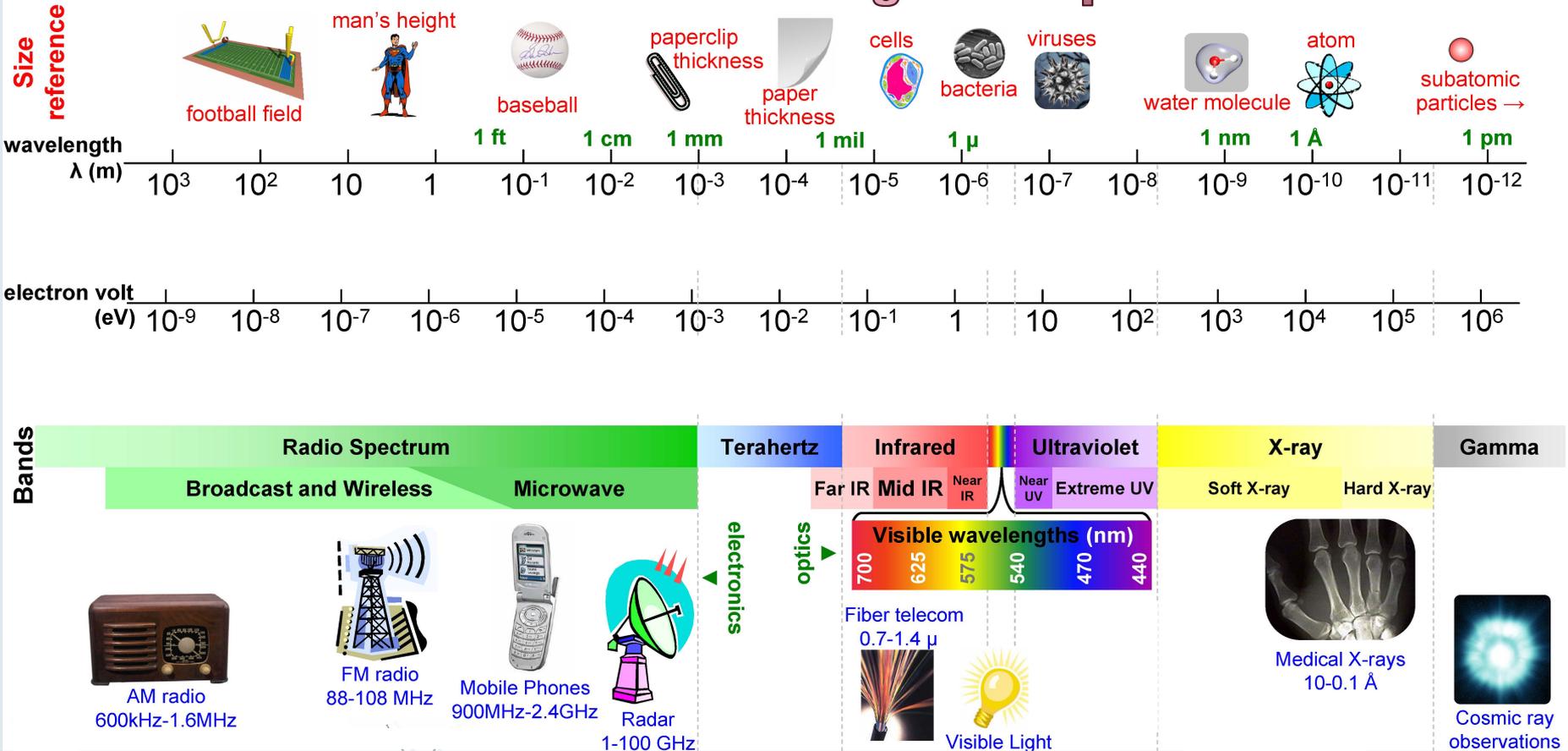
# 1.) Beam-specimen interactions



Basis: Atom model (N.Bohr)

# 1.) Beam-specimen interactions

## Chart of the Electromagnetic Spectrum



$$\lambda \text{ (nm)} = 1.24 / E \text{ (keV)}$$

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# Advantage of electron beam microanalysis

Chemical analysis of a very small volume of material can be done - ideal method for characterisation of a microstructure in a sample !

## Example:

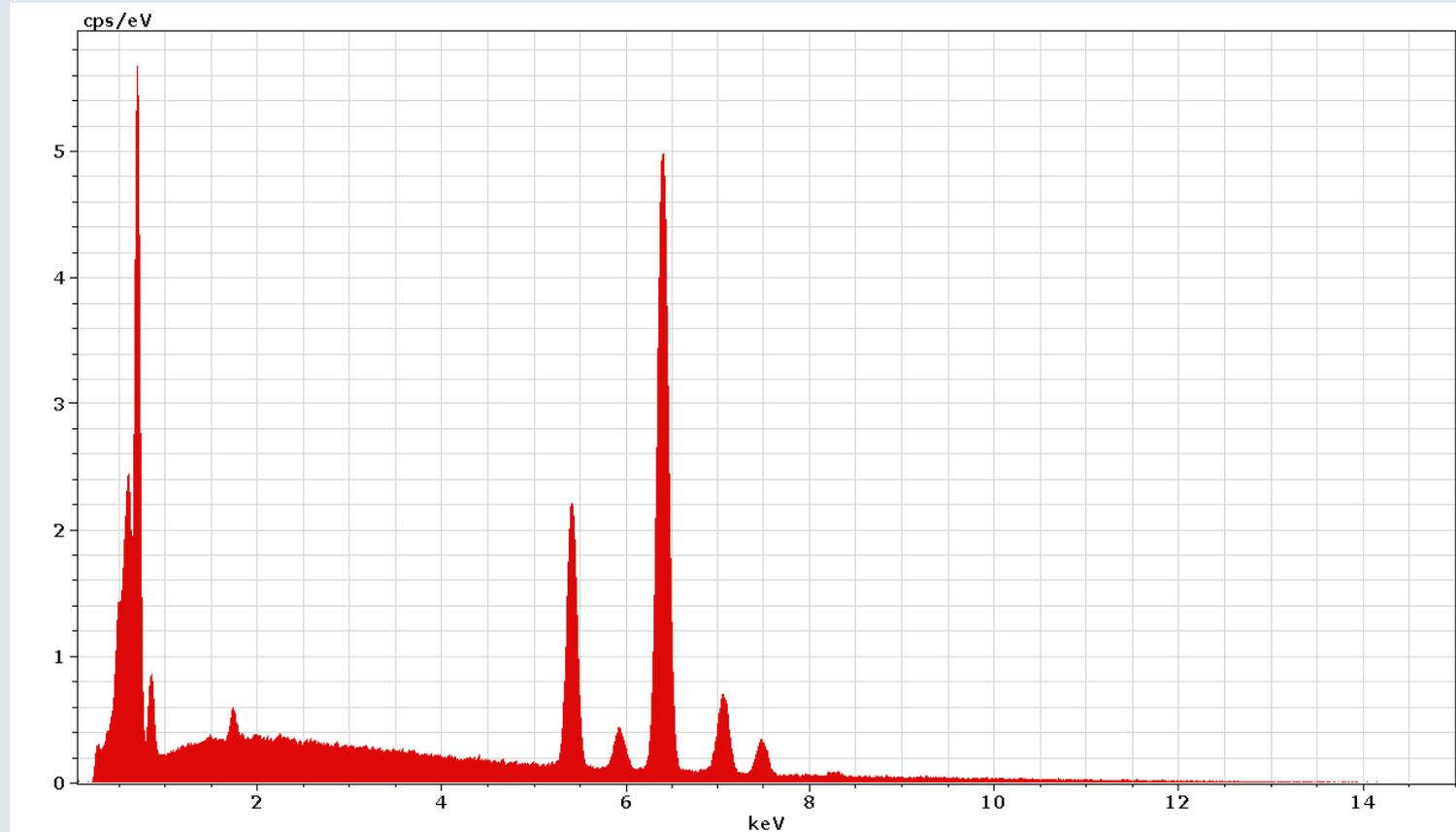
Linear dimension	Analysed volume	Assumed density	Amount of material
<b>1 <math>\mu\text{m}</math></b>	<b><math>\rightarrow 10^{-12} \text{ cm}^3</math></b>	<b><math>\rightarrow 7 \text{ g/cm}^3</math></b>	<b><math>\rightarrow 7 \times 10^{-12} \text{ g}</math></b>

Detection limit: 0,1%

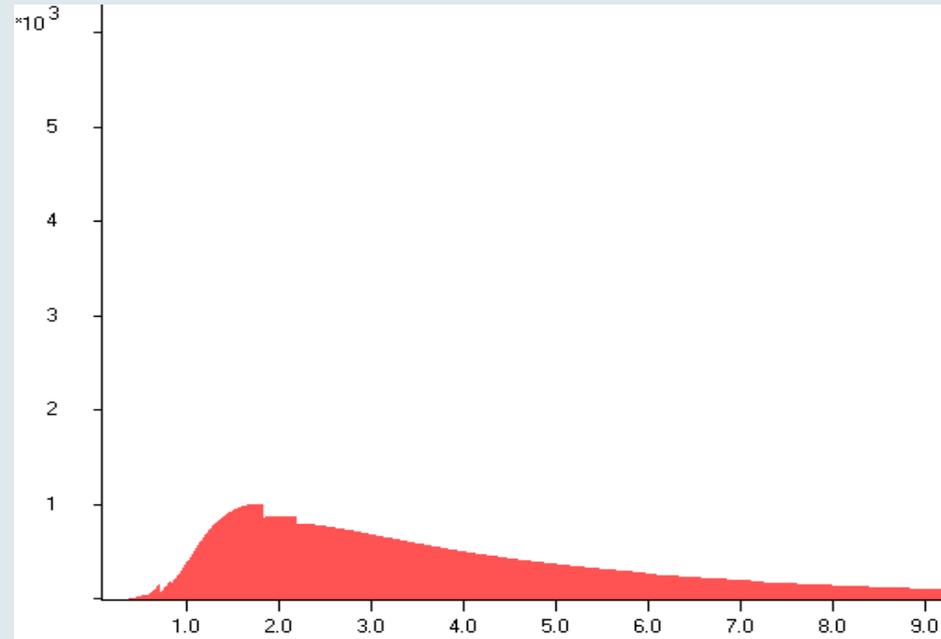
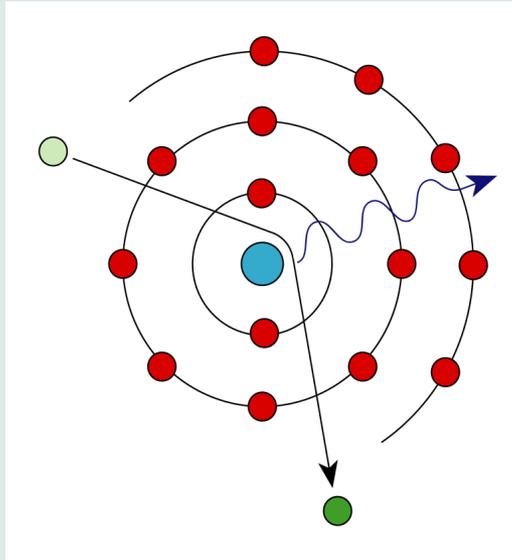
Mass detection limit:  $10^{-14} \text{ g}$

(For reference: Fe - atom weight is about  $10^{-22} \text{ g}$ )

## 2.) EDS spectra: Origin of Bremsstrahlung and characteristic peaks



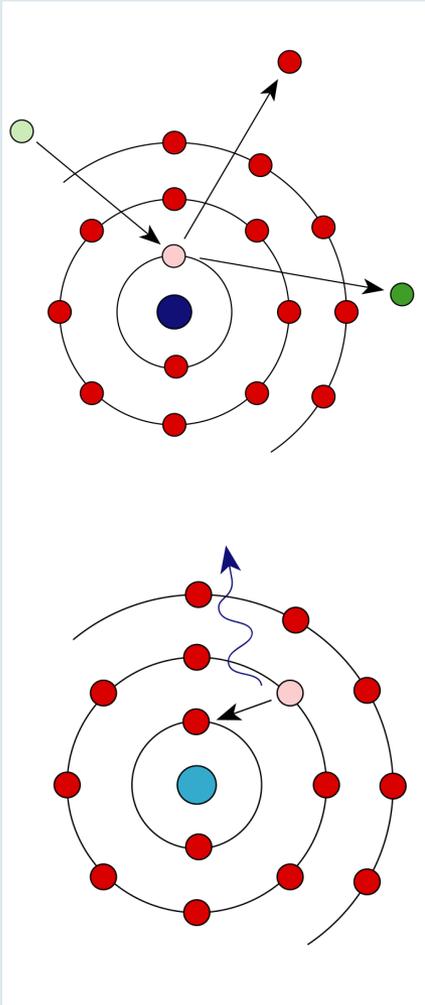
## 2.) EDS spectra: Origin of Bremsstrahlung and characteristic peaks



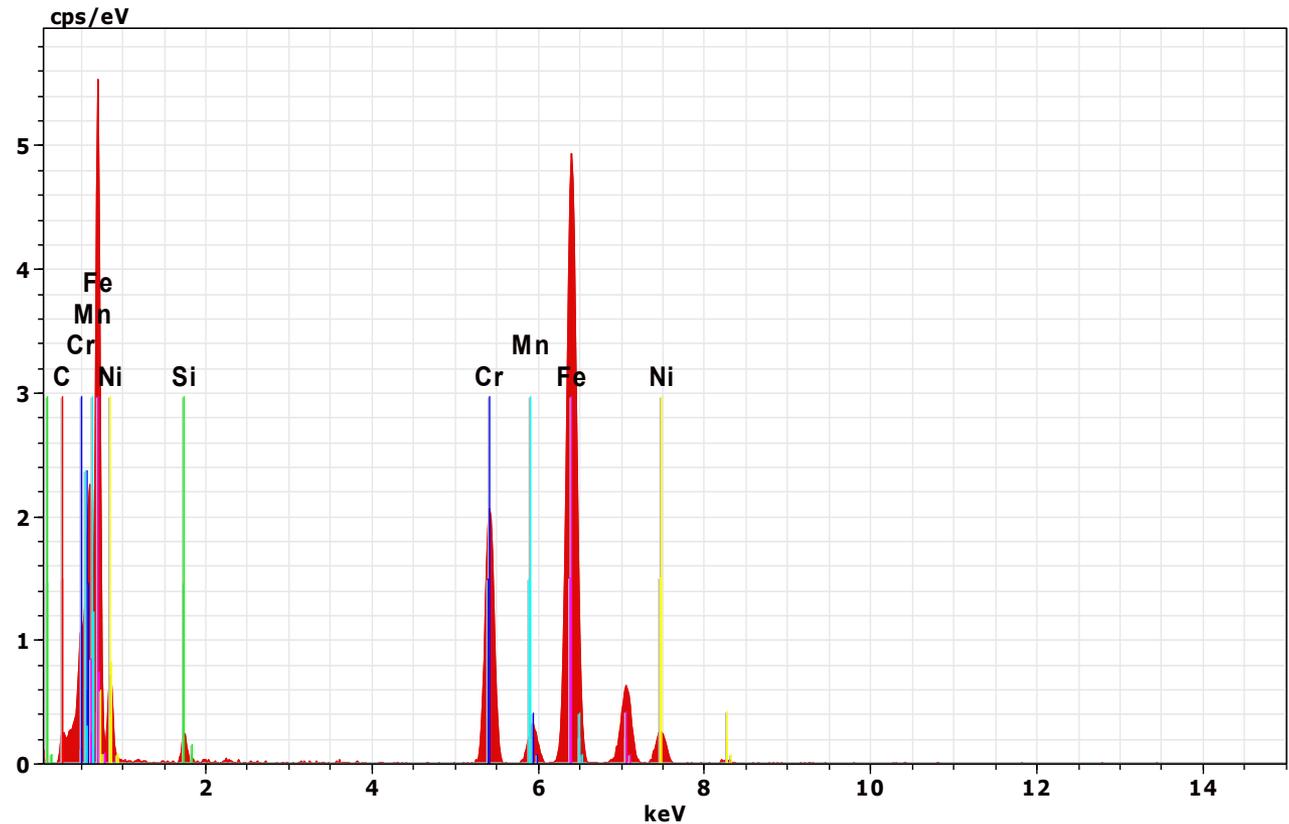
continuum or Bremsstrahlung (braking radiation)

- results from deceleration of beam electrons in the electromagnetic field of the atom core
- combined with energy loss and creation of an X-ray with the same energy

## 2.) EDS spectra: Origin of Bremsstrahlung and characteristic peaks



- Characteristic X-rays are formed by excitation of inner shell electrons
- Inner shell electron is ejected and an outer shell electron replaces it
- Energy difference is released as an X-ray



## 2.) EDS spectra: Origin of Bremsstrahlung and characteristic peaks

If beam energy  $E > E_K$  then a K-electron may be excited



Energy of emitted photon can be calculated:

$$E_{\text{Phot}} = E_1 - E_2$$

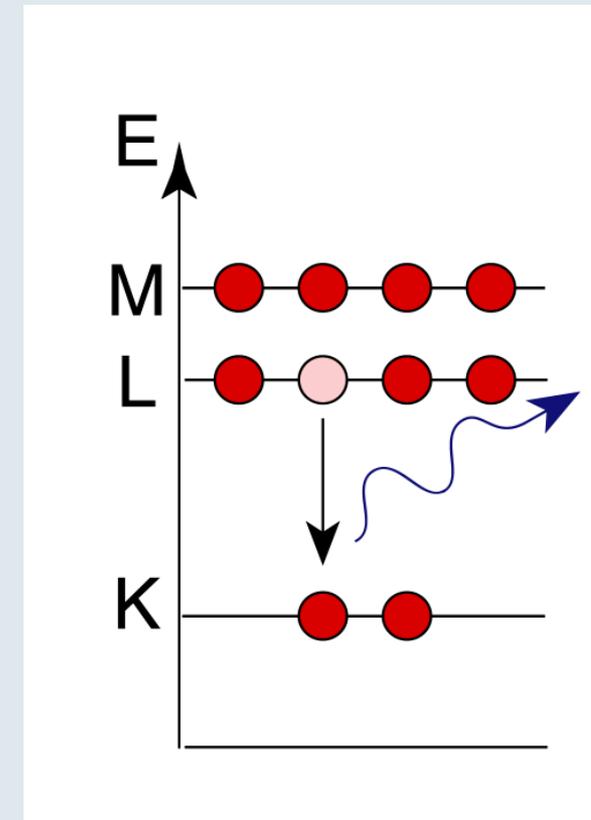
→ X-ray energy is the difference between two energy levels !

e.g.: Fe  $L \rightarrow K$

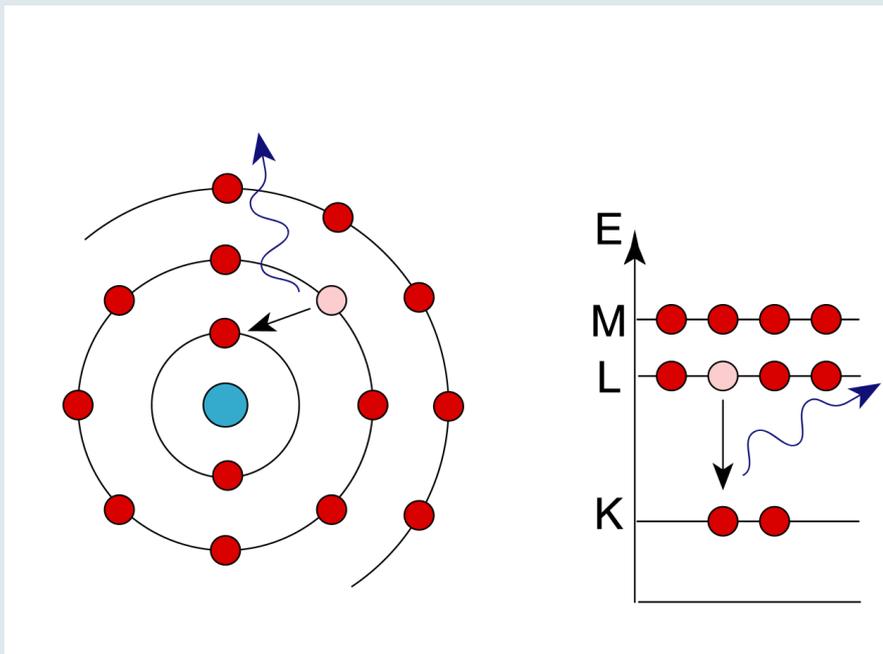
$$E_1 = E_K = 7.11 \text{ keV}$$

$$E_2 = E_L = 0.71 \text{ keV}$$

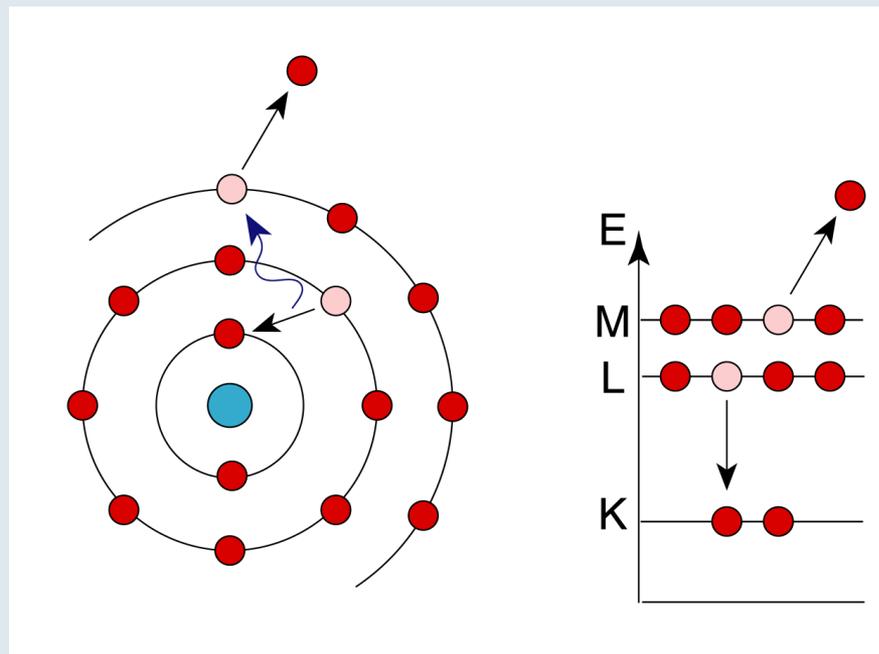
$$E_{\text{Ka}} = 6.40 \text{ keV}$$



# X-ray and AUGER generation process



Emission of X-ray

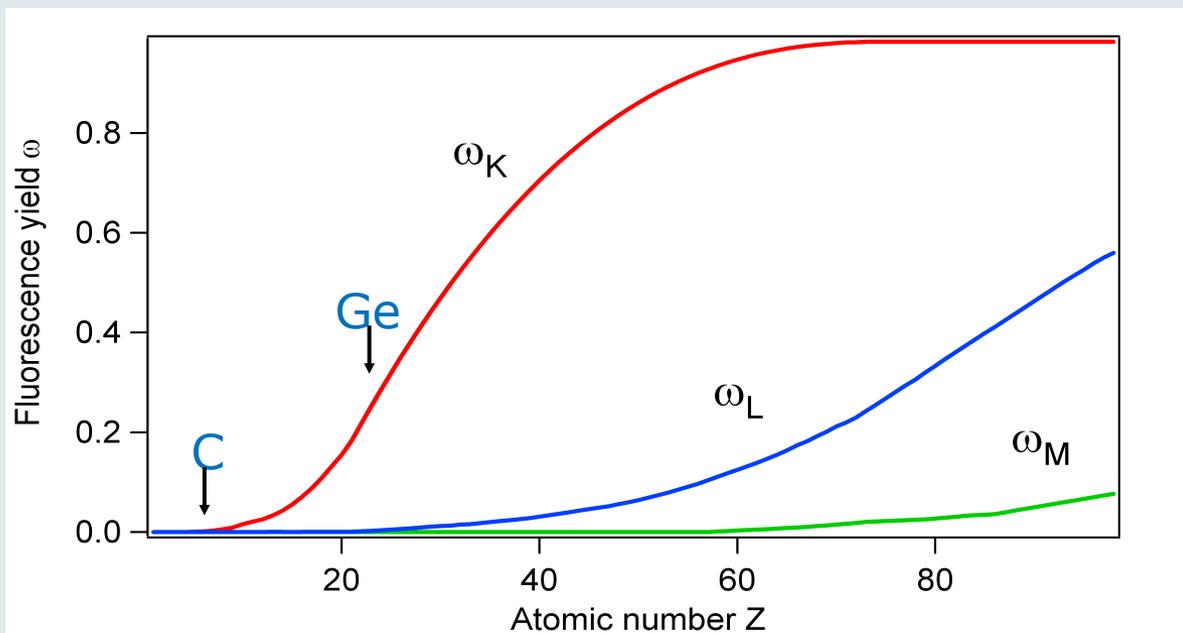


Emission of Auger electron

→ Auger and X-ray yield are competing processes

# Fluorescence yield ( $\omega$ )

- $\omega$  = fraction of ionisation events producing characteristic X-rays (rest produce Auger electrons)



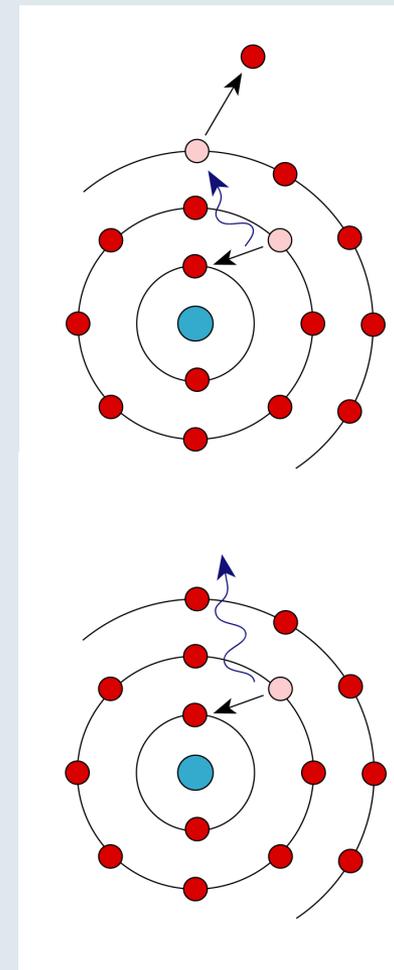
- $\omega$  increases with  $Z$
- $\omega$  for each shell:  $\omega_K$   $\omega_L$   $\omega_M$
- Auger process is favoured for low  $Z$ ,
- fluorescence dominates for high  $Z$



$$\omega + A = 1$$

$$\omega \approx 0.005 \text{ for C K}$$

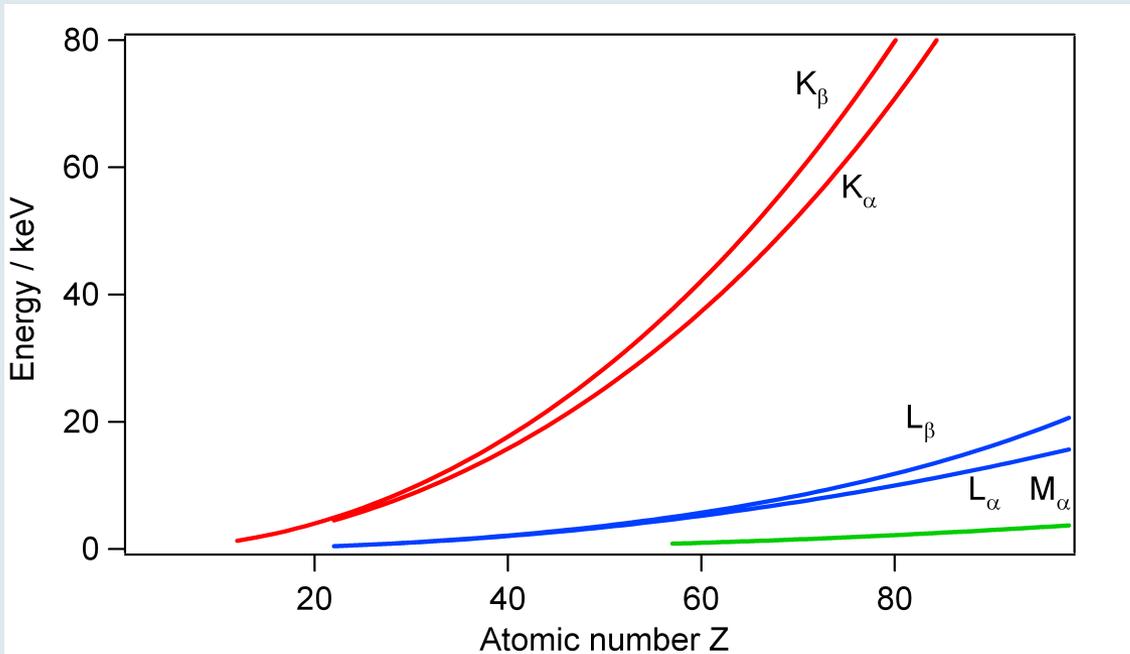
$$\omega \approx 0.5 \text{ for Ge K}$$



# Outline

- 1.) Beam-specimen interactions
- 2.) EDS spectra: Origin of Bremsstrahlung and characteristic peaks
- 3.) Moseley's law
- 4.) Characteristic peaks: K-, L-, and M series
- 5.) Spatial resolution and excitation range in EDS analysis

# 3.) Moseley's law

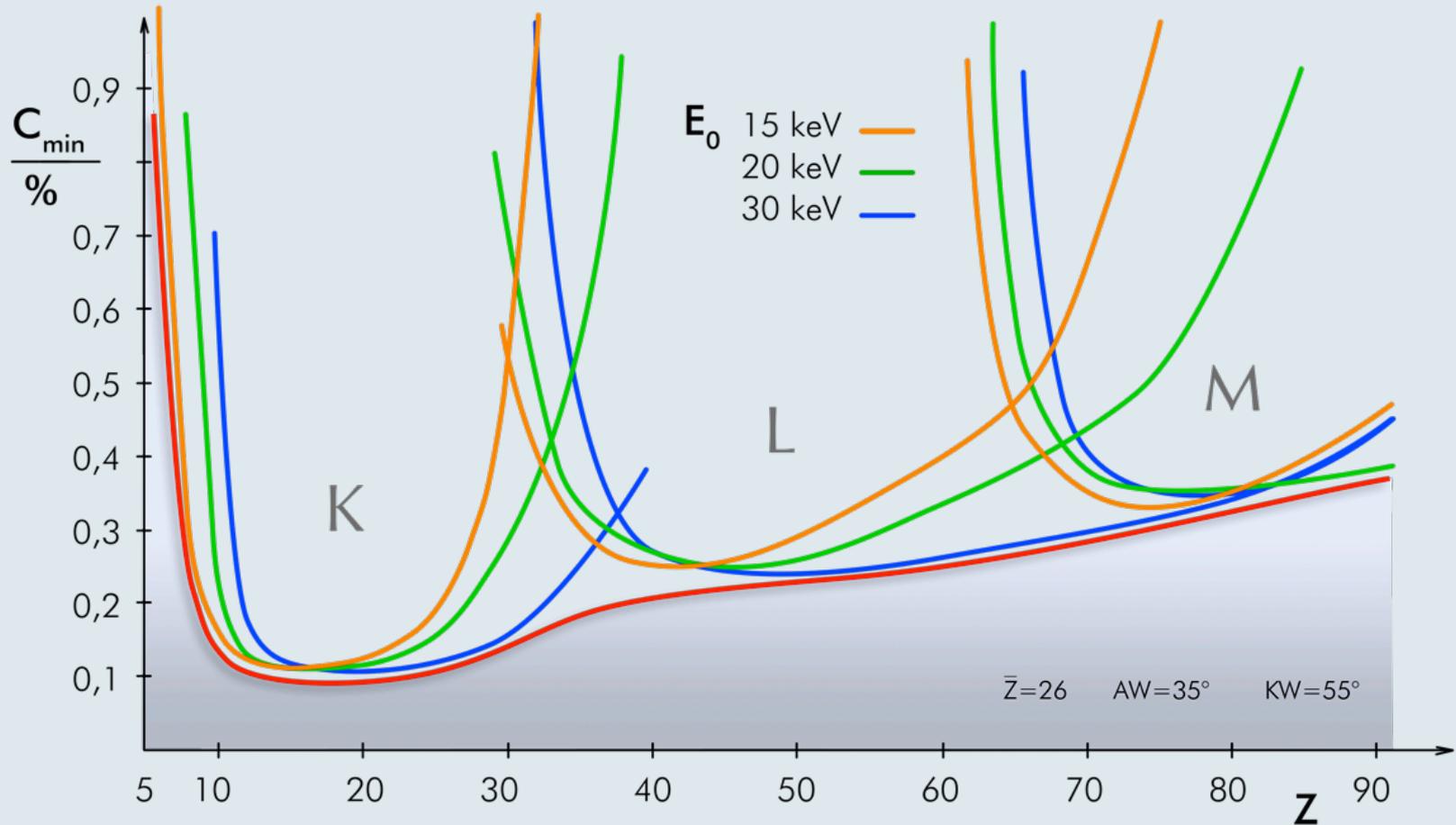


X-rays are **characteristic** because their specific energies are characteristic of the particular element which is excited.

$$E = c_1 (Z - c_2)^2$$

Moseley's law defines the relationship between the x-ray lines and the atomic number of the emitted atom.

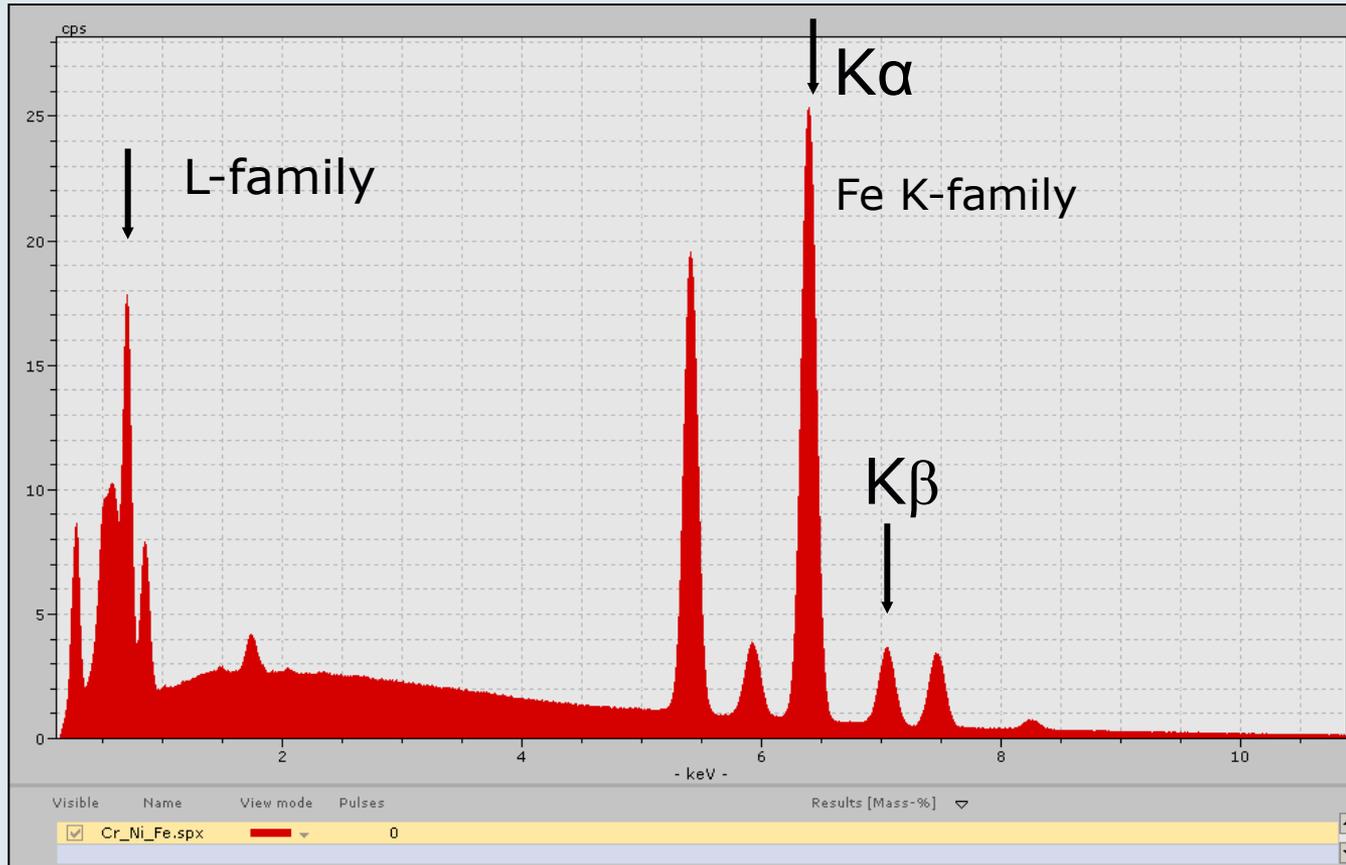
# Detection limit of EDS



# Outline

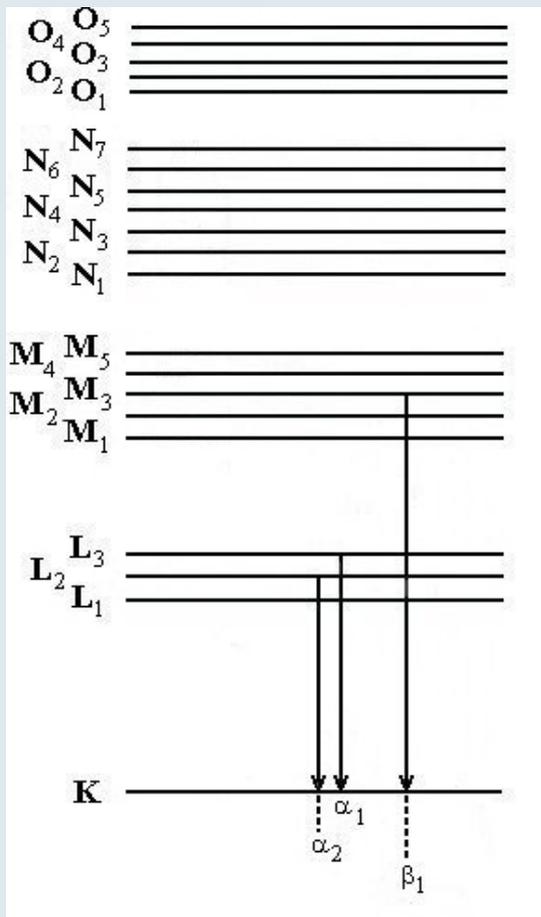
- 1.) Beam-specimen interactions
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## 4.) Characteristic peaks: K, L, M series



- Energy of characteristic peaks is defined by element
- The higher the atomic number  $Z$  the higher the peak energy

# The K-family of lines (1)

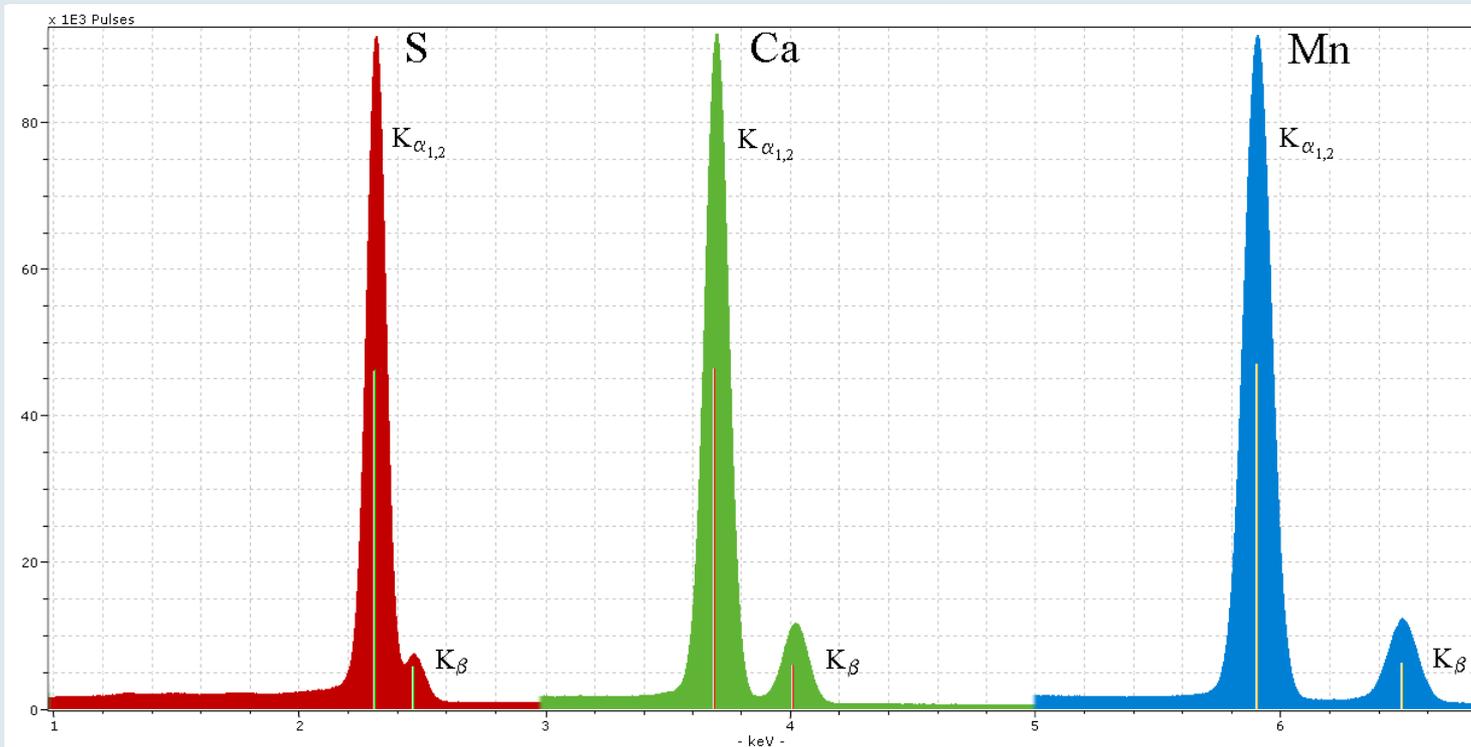


K-lines: vacancy in K-level is filled  
 $\alpha$ - lines are L -K transitions  
 $\beta$ - lines are M-K transitions

## The K-family of lines (2)

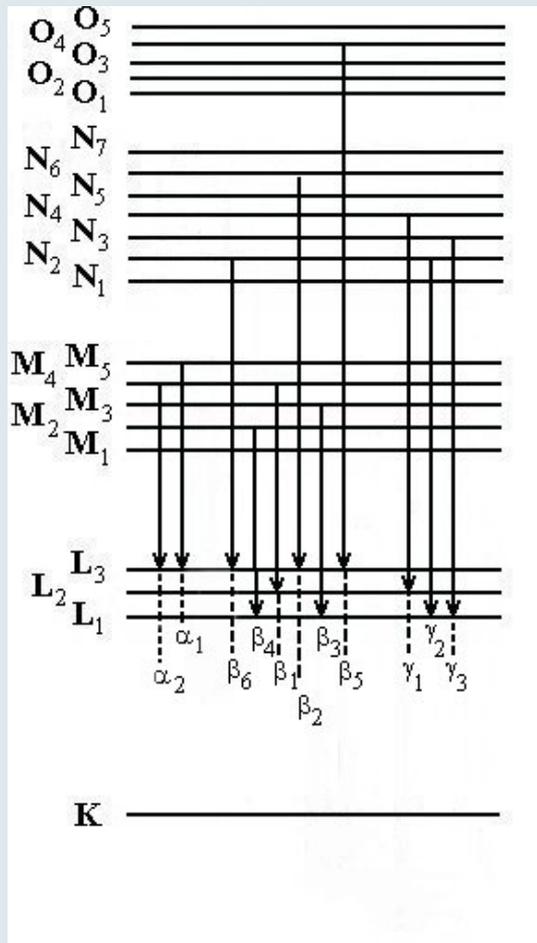
- K lines in ED spectra are either a combination of  $K\alpha + K\beta$  peaks or a separated pair ( $K\beta$  weight then about  $1/8 \dots 1/10$ )
- Below element S ( $K\alpha = 2308$  eV) it is not possible to resolve the two peaks with EDS  $\rightarrow$  a  $K\beta$  shoulder may be visible on the high-energy side of the  $K\alpha$
- line energy difference ( $K\alpha$ - $K\beta$ ) is increasing with atomic number
- for SEM (30 kV  $U_{\max}$ ) K lines up to atomic number 42 (Mo) can be excited

# The K-family of lines (3)



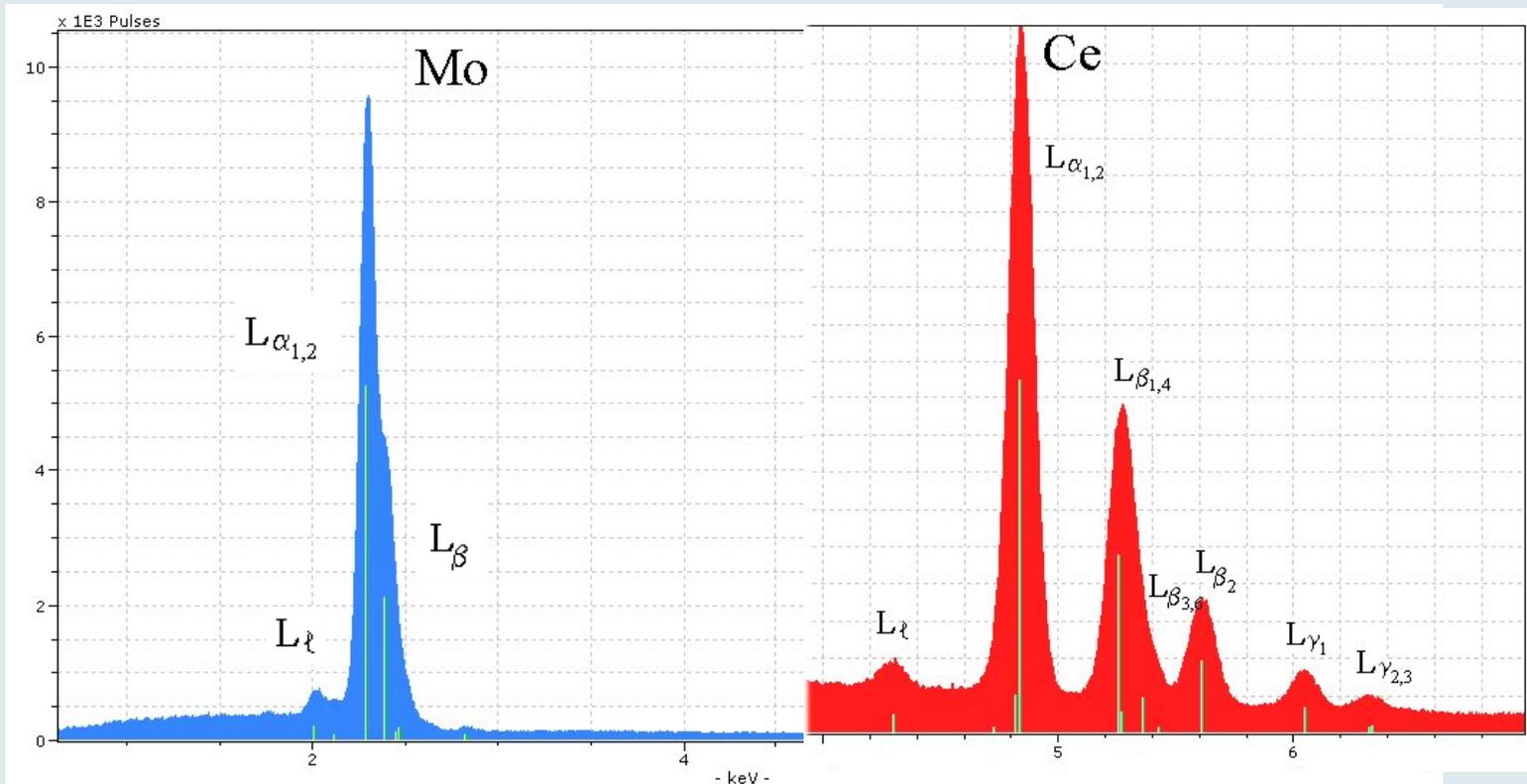
	S (16)	Ca (20)	Mn (26)
$K\alpha_{1,2}$	2308 eV	3692 eV	5900 eV
$K\beta$	2464 eV	4013 eV	6492 eV
$\Delta (K\beta - K\alpha_{1,2})$	156 eV	319 eV	592 eV

# The L-family of lines (1)



- L-lines occur: vacancy in L-level is filled

# The L-family of lines (2)

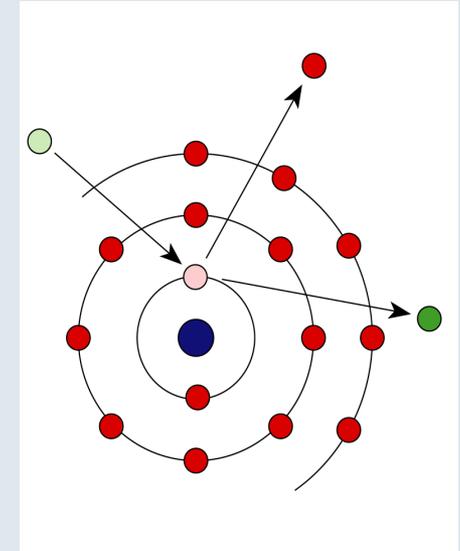
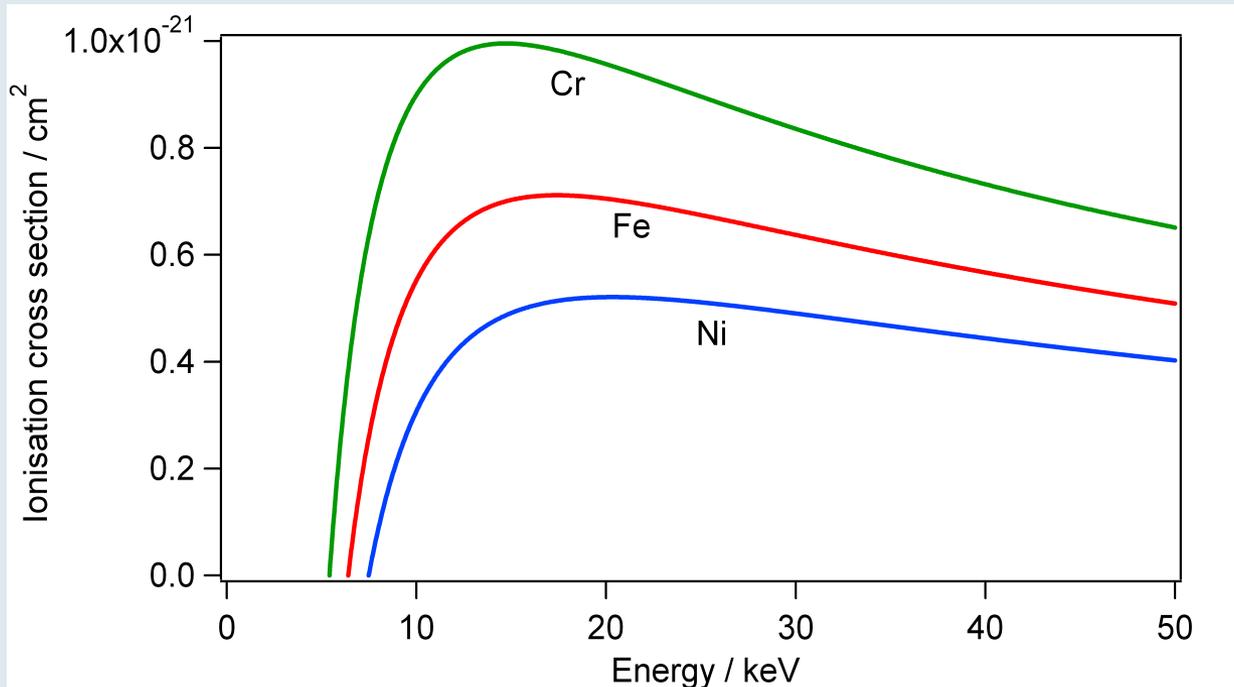


	Mo (42)		Ce (58)
$L_{\alpha_1}$	2292 eV		4839 eV
$L_{\beta}$	2394 eV		5262 eV
$L_1$	2014 eV		4287 eV

# Intensity and energy of characteristic lines

- Energy of line is defined by
  - Element
  - Type of transition
  
- Intensity of line is defined by
  - probability of producing a hole (vacancy)
  - probability of electron transition
  - probability of x-ray emission
  - concentration

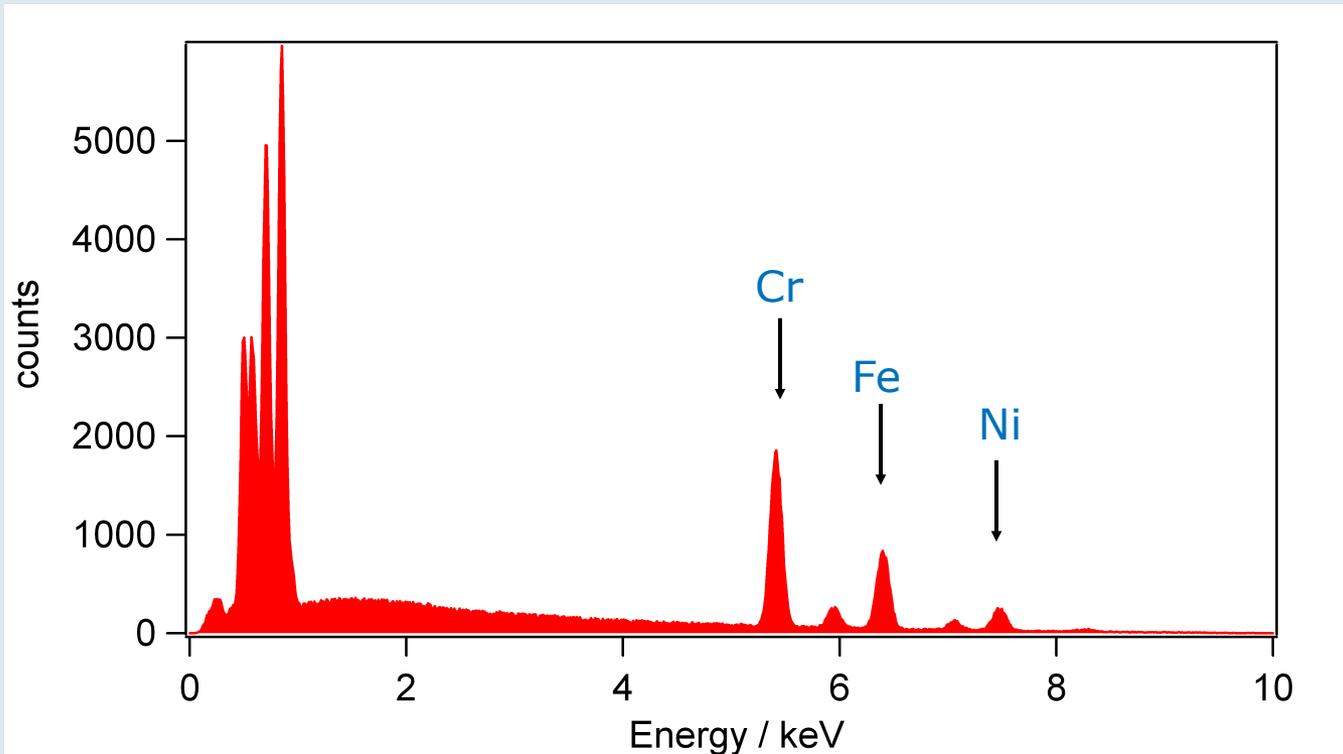
# Probability of producing a hole: Ionization cross-section for electrons



Ionisation cross section for electrons

- Ionization cross section: probability of excitation
- maximum ionization cross section:  $2,5 \times E_{\text{bind}}$

# Ionization cross section for electrons



Cr: 33%

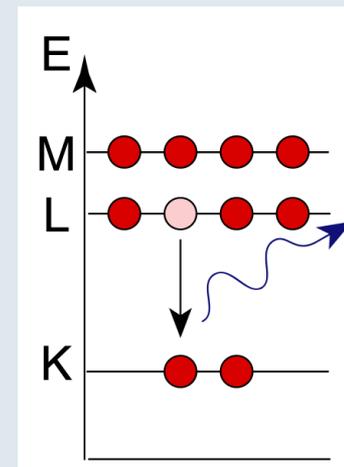
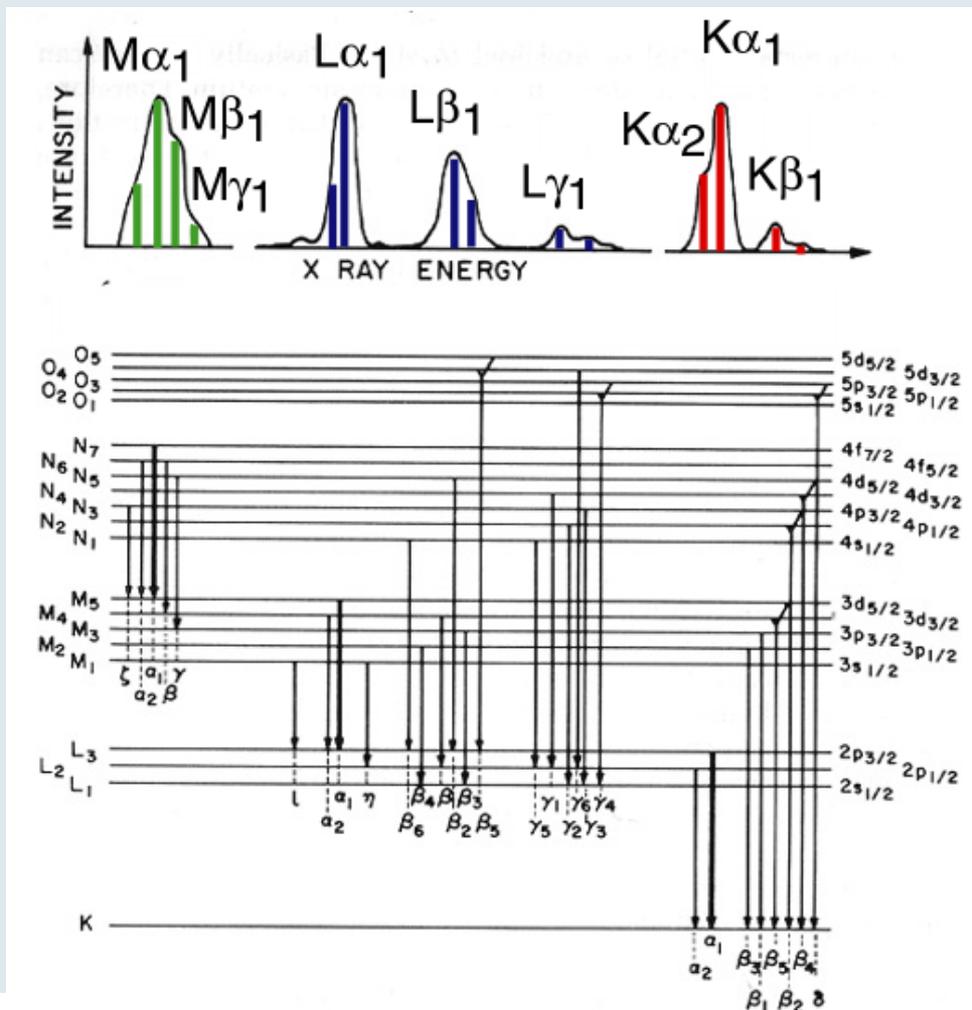
Fe: 33%

Ni: 33%

U = 10 keV

$\frac{E_{exc}}{E_{bind}}$ :	Cr	Fe	Ni
	1,847	1,561	1,337

# Atomic energy levels and line transition

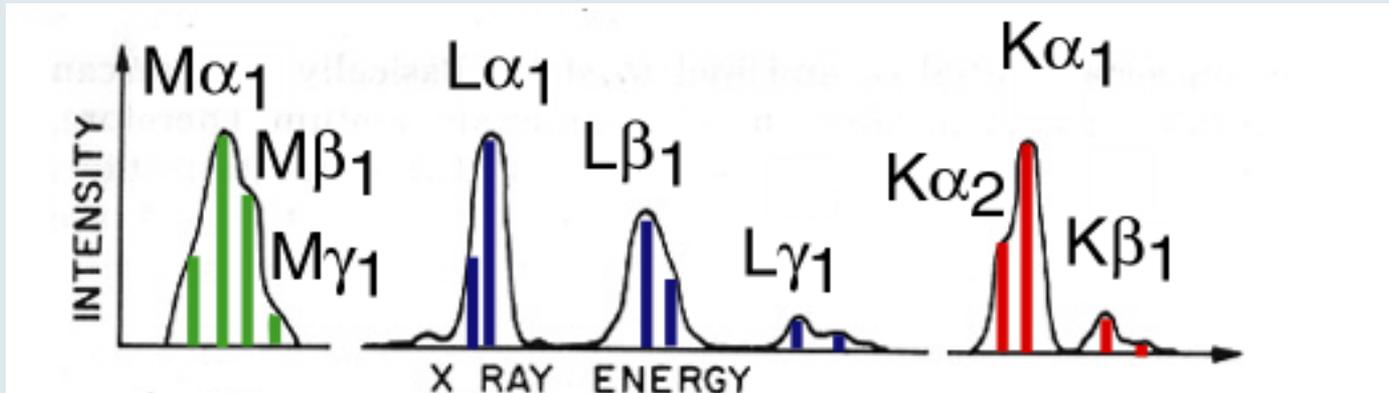


Transitions have different probabilities



Lines have different intensities

# Line intensity relations



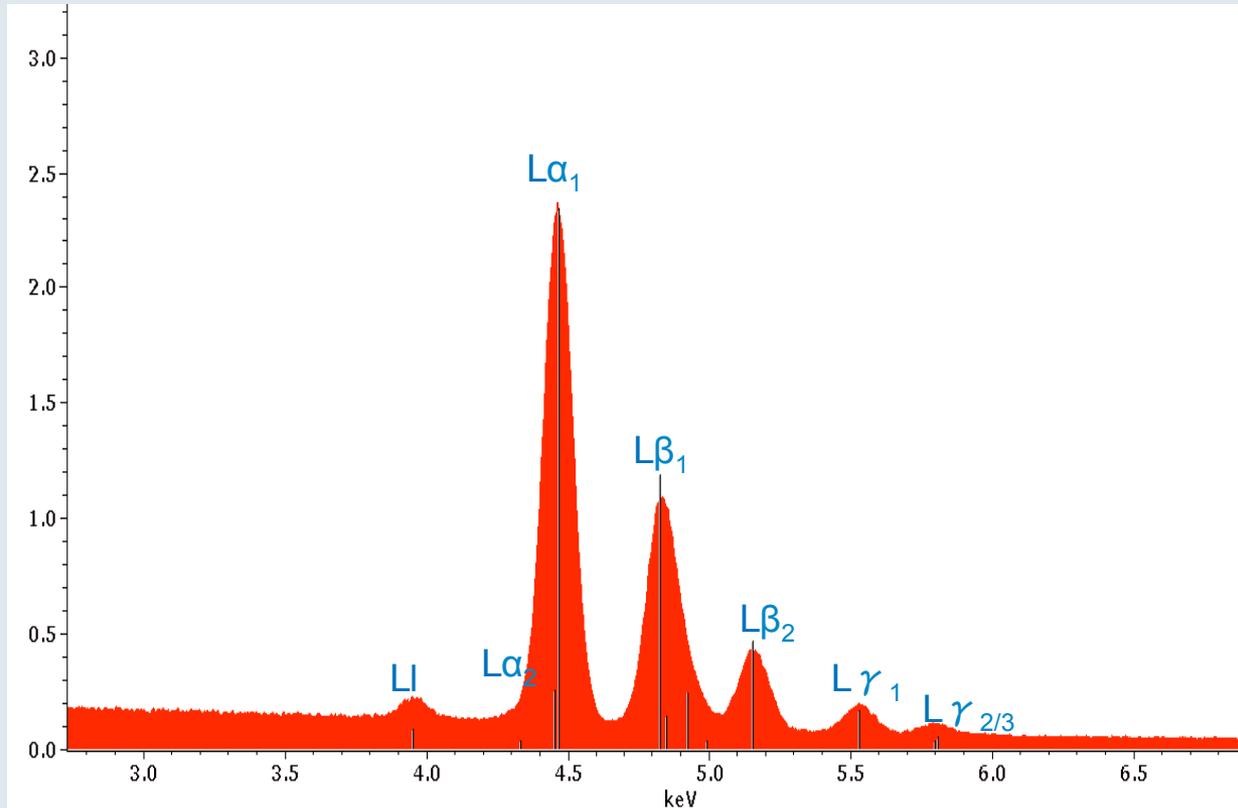
Intensity of an x-ray line is determined by the transition probability of electrons from the outer to inner shell. These values are fixed for the lines of one series.

K- series:  $Z < 12 \rightarrow \alpha_1 : \alpha_2 : \beta_1 = 100 : 50 : 15$

$13 > Z < 50 \rightarrow \alpha_1 : \alpha_2 : \beta_1 = 100 : 53 : 18$

L- series:  $15 > Z < 90 \rightarrow \alpha_1 : \beta_1 : \gamma_1 = 100 : 52 : 10$

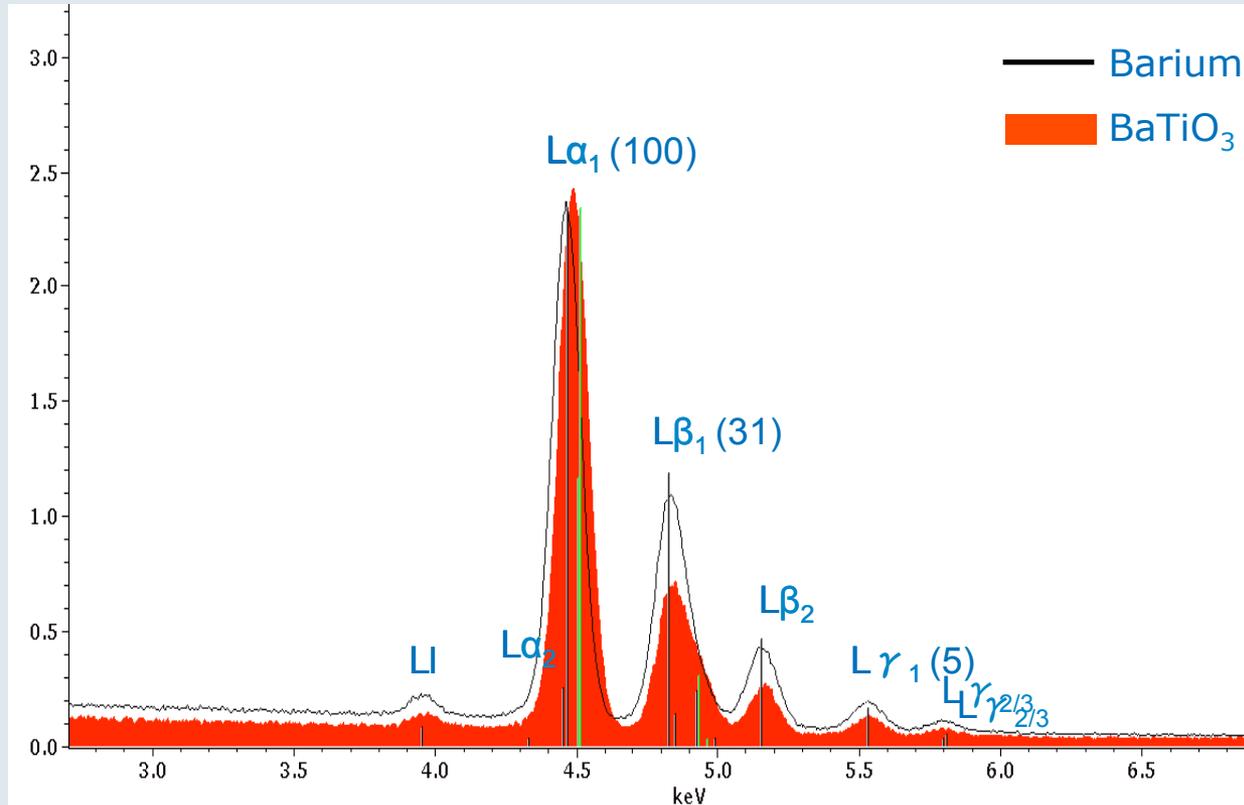
# Line intensity relations (2)



Spectrum Barium L series, 15 kV

$\alpha_1 : \beta_1 : \gamma_1 = 100 : 52 : 10$

# Line intensity relations (3)



~~Spectrum Barium L-series, 15 kV~~

~~$\alpha_1 : \beta_1 : \gamma_1 = 100 : 52 : 10$~~

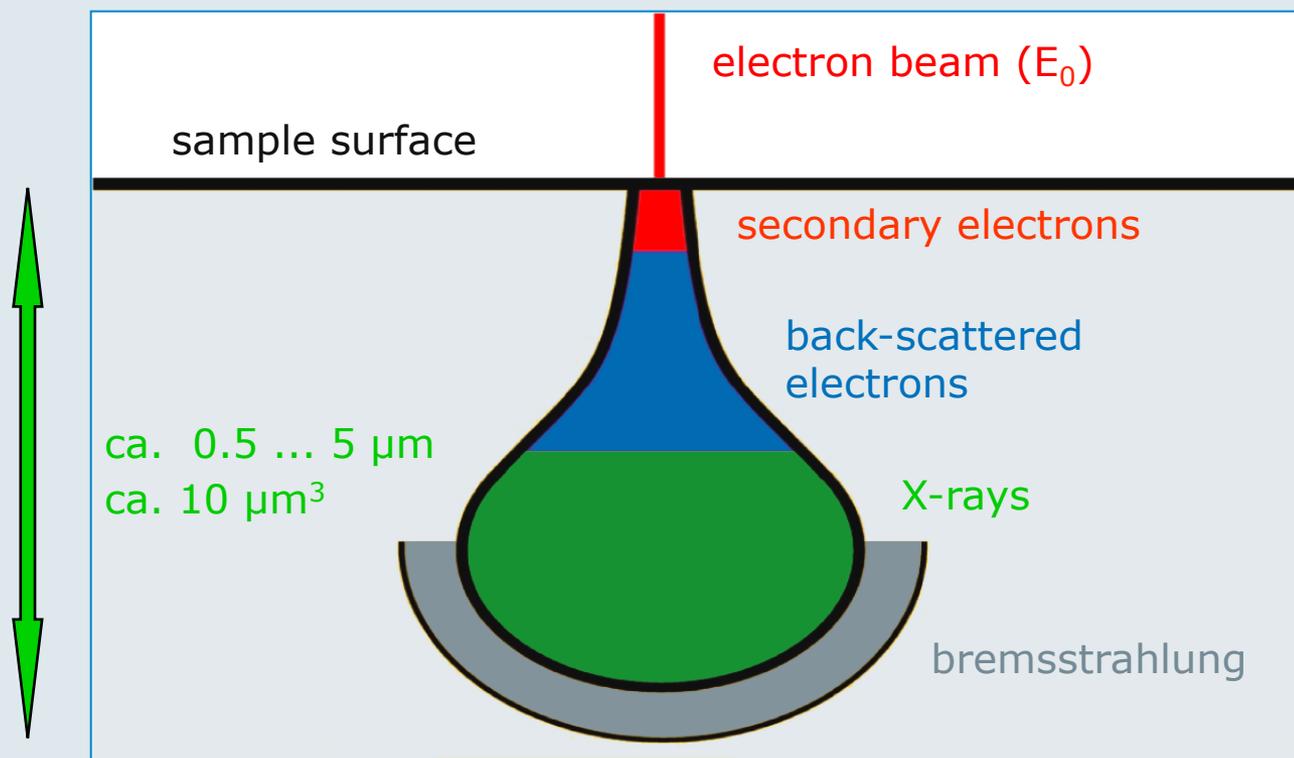
Spectrum BaTiO<sub>3</sub>, 15 kV

Overlapped Ba L-series and Ti K-series

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## 5.) X-ray range



- Different excitation ranges for:
- characteristic x-ray radiation and Bremsstrahlung,
  - secondary electrons (SE)
  - back-scattered electrons (BSE)

## 5.) X-ray range

Dependence HV - x-ray range:

Anderson and Hasler (1966) give the depth of X-ray production range ( $\mu\text{m}$ ) as:

$$R_{AH} = 0.064(E_0^{1.68} - E_c^{1.68}) / \rho$$

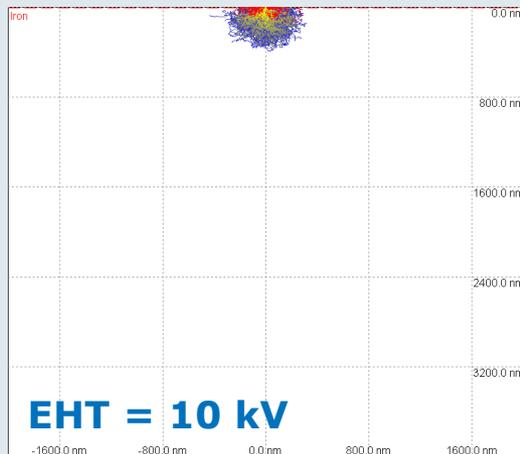
$E_0$ : primary energy (keV),

$E_c$ : critical energy (keV),

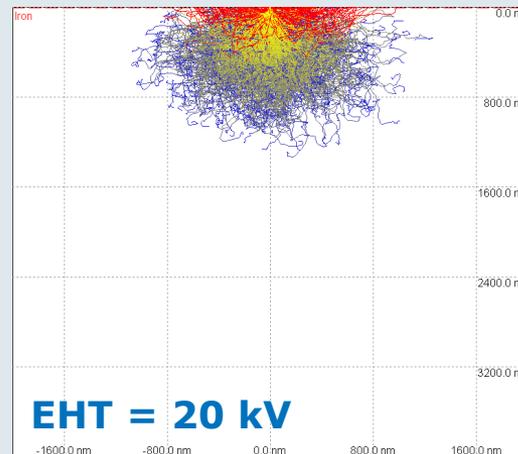
$\rho$ : mean density ( $\text{g}/\text{cm}^3$ )

## 5.) X-ray range

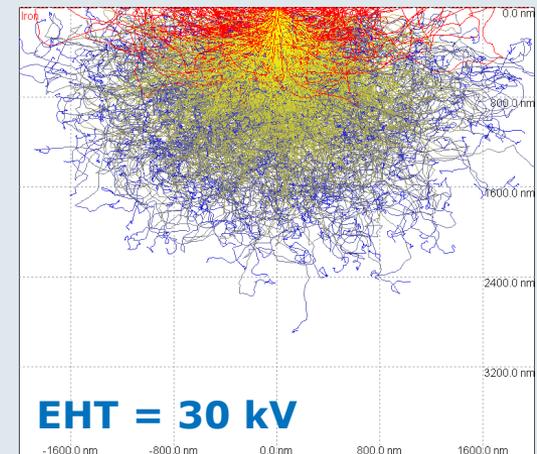
- Monte Carlo electron-trajectory simulations of interaction volume in iron as function of primary beam energy



$$R_d \approx 0,4 \mu\text{m}$$



$$R_d \approx 1,3 \mu\text{m}$$

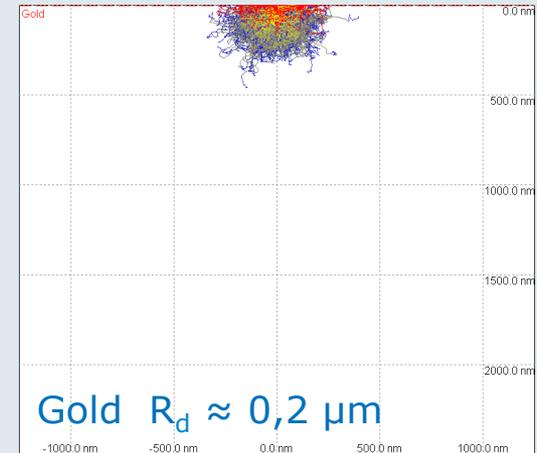
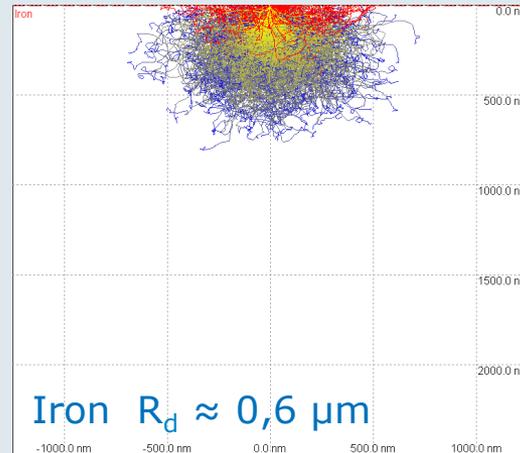
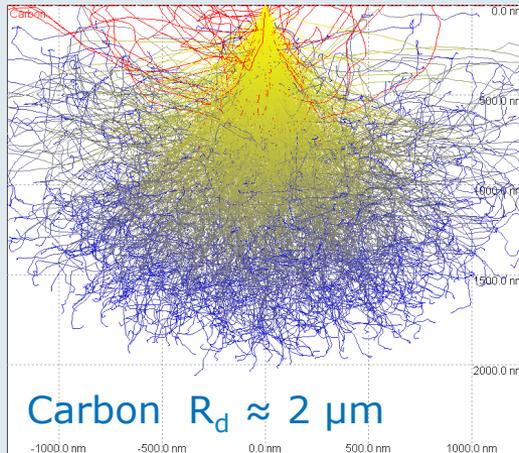


$$R_d \approx 2,5 \mu\text{m}$$

→ With higher primary electron energy penetration depth is increasing

# 5.) X-ray range

- Monte Carlo electron-trajectory simulations of interaction volume as function of atomic number (EHT = 15 kV)



→ With higher density penetration depth is decreasing

# The kV compromise

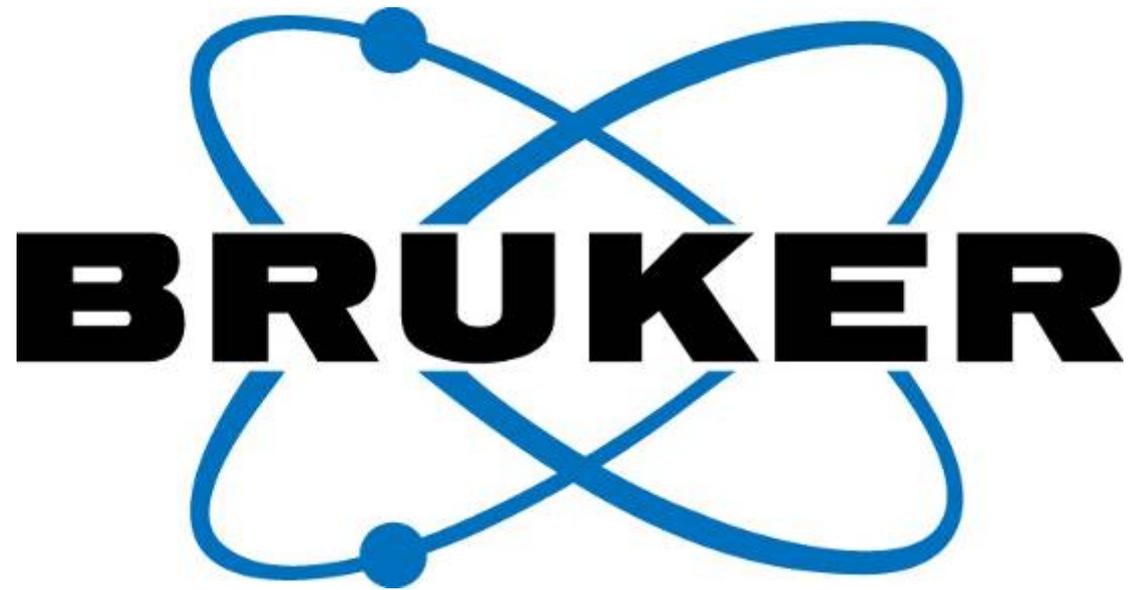
$I_{\text{char}}$  increases with increasing  $E_0/E_c$

→ X-ray signal improves

$R_x$  increases with increasing  $E_0/E_c$

→ X-ray spatial resolution degrades

$$U_0 = \frac{E_0}{E_c} > 2 \dots 2,5$$



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